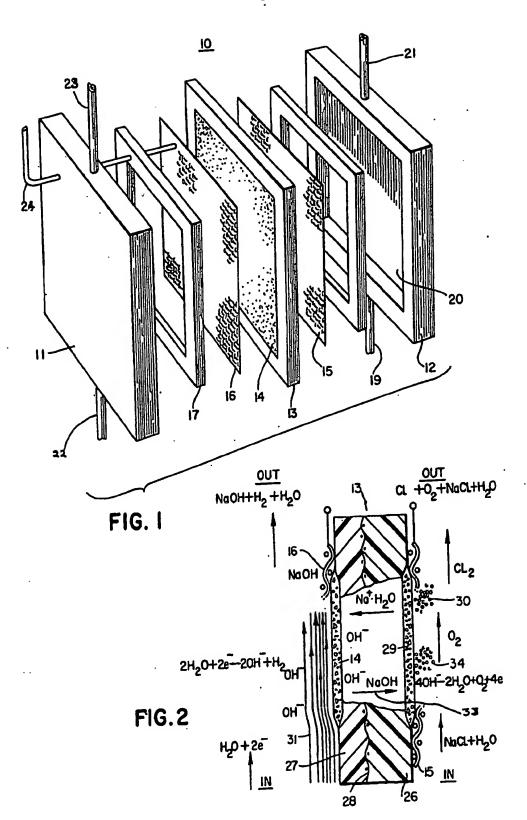
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- (54) Thermally stabilized, reduced platinum metal oxide electrocatalyst
- (57) Electrocatalytic material comprising at least one reduced platinum group metal oxide is subsequently heated in the presence of oxygen at a temperature high enough to stabilize the catalyst in acidic and halogen environments. The catalyst optionally contains thermally stabilized oxides of transition or valve metals, nitrides, carbides or sulphides thereof, and graphite. A gas permeable electrode structure includes particles of the catalyst and a polymeric binder. A method of preparing the electrocatalytic material is described as well as a unitary electrolyte electrode structure which has a gas permeable electrode containing thermally stabilized particles of the electroca-

talytic material, bonded to at least one side of a membrane-electrolyte.



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## **SPECIFICATION**

## Thermally stabilized, reduced platinum metal oxide electrocatalyst

5 The instant invention relates to a electrocatalyst, a catalytic electrode, and a membrane/electrode assembly. More particularly, it relates to catalysts and electrodes which are particularly useful in the electrolysis of halides. Generating gas by electrolyzing a chemical compound into its constituent elements, one of which may be a gas, is, of course, an old and well known technique. One recently developed 10 form of such gas evolving electrolyzer involves the use of a cell which utilizes an electrolyte in 10 the form of a solid polymer, ion-exchanging membrane. In an arrangement of this sort, catalytic electrodes using a suitable catalyst are positioned on opposite sides of an ion transporting membrane medium such as a sulfonated perfluorocarbon ion-exchange membrane. Through an oxidation reaction, the ionic form of one of the constituent elments (hydrogen ions, for example, 15 when H<sub>2</sub>O or HCl is electrolyzed, or sodium ions when an alkali metal halide such as sodium 15 chloride is electrolyzed) is produced at one electrode. The ion is transported across the ionexchanging membrane to the other electrode where it is reduced to form an electrolysis product such as molecular hydrogen, NaOH, etc. Solid polymer ion-exchange membranes electrolysis units are particularly advantageous because they are efficient, small in size, and do not utilize 20 20 any corrosive liquid electrolytes. Various metal and alloys have been utilized in the past as part of the catalytic electrodes associated with such electrochemical electrolyzing cells. The performance of the catalyst at the gas evolving electrodes is obviously crucial in determining the effectiveness and efficiency of the cell, and consequently of the economics of the process. The choice of a catalyst in an 25 electrochemical cell and its effectiveness depends upon a complex set of variables, such as 25 surface area of a catalyst, availability of oxides of its species on the catalyst surface, contaminants in the reactants, and the nature of the conversion taking place in the cell. Consequently, it is, and always has been, difficult to predict the applicability of a catalyst useful in one electrochemical cell to a different system. A commonly assigned Patent No. 3,992,271 30 30 entitled "Methods and Apparatus for Gas Generation" describes an improved oxygen evolving catalytic electrode utilizing a platinum-iridium alloy, a mixture which was found to provide much improved performance and efficiency. Another commonly assigned U.S. Patent No. 4,039,490 describes another oxygen evolving catalytic electrode which utilizes reduced oxides of platinumruthenium. The platinum-ruthenium catalyst not only is substantially less expensive than the 35 reduced platinum-iridium catalyst, because it uses a less expensive material such as ruthenium to alloy with the platinum, but it also turns out to be more efficient because it has a lower oxygen overvoltage than a platinum-iridium electrode. However, attempts to use reduced ruthenium oxide electro-catalysts for evolution of halogens by electrolysis of aqueous halide solutions have not been entirely successful due to the harsh 40 40 electrolysis conditions in the cell. There can be substantial loss of catalyst from the membrane during chlorine evolution since these reduced platinum metal oxides are susceptible to dissolution in acidic environments which are present in the electrolysis of hydrogen halides or in the electrolysis of alkali metal halide solutions which are often acidified. Not only is there a tendency to dissolution of the platinum metals resulting in a loss of a catalytic material, but the 45 overvoltage of the electrodes also tends to increase so that the efficiency of the cell decreases, 45 and in many instances does not permit prolonged periods of operation.

The advantages of the invention will become apparent as the description thereof proceeds. In accordance with the invention, the novel electrocatalyst comprises at least one reduced platinum group metal oxide which is subsequently treated in the presence of oxygen at a 50 temperature high enough to stabilize the oxide thermally to increase the resistance of the catalyst against the corrosive electrolysis conditions. The catalytic, reduced platinum group metal oxide may optionally contain other reduced platinum group metal oxides such as iridium and optionally up to fifty (50) percent by weight of the electroconductive extenders such as graphite, valve metal oxides, titanium metal oxides and nitrides, carbides, and sulfides. 55 Examples of useful platinum group metals are platinum, palladium, iridium, rhodium, ruthenium, and osmium with the preferred reduced metal oxide for chlorine and other halogen production being thermally stabilized, reduced oxides of ruthenium. Reduced oxides of ruthenium are preferred because they are found to have extremely low chlorine overvoltages as well as

their stability in the electrolysis environment. As pointed out above, the electrocatalytic material may be a single reduced platinum group metal oxide such as ruthenium oxide, or platinum oxide, or iridium oxide, etc. It has been found, however, that mixtures or alloys of thermally stabilized, reduced platinum group metal avides are even more etable. One such mixture or alloy of ruthenium oxide containing Up to

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than ruthenium alone.

Electroconductive extenders such as graphite have low over-voltages for halogens and are substantially less expensive than the platinum metal oxides and may readily be incorporated without reducing the effectiveness of the catalyst. In addition to graphite, oxides of valve metal such as titenium, tantalum, niobium, tungsten, vanadium, zirconium, and hafnium may be added to further stabilize the electrocatalyst and increase its resistance against adverse electrolysis conditions.

The thermally stabilized, reduced platinum metal oxides and the extenders thereto formed into an electrode by bonding with fluorocarbon resin particles such as those sold by Dupont under its trade designation Teflon. The catalytic particles and resin particles are mixed, placed in a mold and heated until the composition is sintered into a suitable form which is bonded to at least one surface of the membrane by application of heat and pressure to provide an electrode structure and a unitary membrane/electrode structure.

The novel process for the preparation of the electrocatalyst comprises forming oxides of at 15 least one platinum group metal along with one or more extenders such as graphite, valve metals, reducing the oxide to a partially oxidized state and then heating the latter in the presence of oxygen at a temperature which is sufficiently high to stabilize the reduced oxides.

The novel features which are believed to be characteristic of this invention are set forth with particularity in the appended claims. The invention itself, however, both as to its organization and method of operation, together with further objects and advantages thereof, may best be understood by reference to the following description taken in conjunction with the accompanying drawings in which:

Figure 1 is a schematic illustration of an electrolysis cell in accordance with the invention utilizing a solid polymer electrolyte membrane and novel catalyst bonded to the surface thereof.

Figure 2 is a schematic illustration showing the reactions taking place in various portions of the cell during electrolysis of an aqueous halide solution.

The novel electrocatalyst which includes thermally stabilized, reduced oxides of a platinum group metal along or in combination with other platinum group metals or optional valve metals may be prepared in any convenient fashion whereby an oxide catalyst is permanent, partially 30 reduced and thermally stabilized.

The preferred manner of reduction is by a modification of the Adams method of platinum preparation by the addition of a thermally decomposable platinum halide, such as ruthenium chloride, either alone or, if desired, along with an appropriate quantity of other thermally decomposable halides of platinum metals or valve metals to an excess of sodium nitrate. The Adams method of platinum preparation is disclosed in an article published in 1923 by R. Adams and R. L. Schriner in the Journal of the American Chemical Society, Volume 45, Page

217. It is convenient to mix the finely divided hallde salts of the platinum metals, such as Chloroplatinic acids in the case of platinum, ruthenium chloride in the case of ruthenium, titanium chloride, tantanium chloride, in the case of titanium and tantalum in the same weight ratio of the metals as desired in the final alloy mixture. An excess of sodium nitrate is incorporated and the mixture is fused in a silica dish at 500 to 600°C for three (3) hours. The residue is washed thoroughly to remove nitrates and halides still present, leaving a residue of

the desired platinum metal oxide, i.e., ruthenium oxide, platinum-ruthenium oxide, rethenium-iridium oxide, ruthenium-titanium oxide, etc. The resulting suspension of mixed and alloyed oxides is then partially reduced. The reduction of the platinum group metal oxides may be effected by any convenient known reducing method, such as an electrochemical reduction or by bubbling hydrogen through the mixture at room temperature as long as the oxides are not be to completely reduced to the free metal form. In a preferred embodiment, oxides are reduced by using an electrochemical reduction technique, i.e., electrochemical reduction in an acid medium.

50 The product which is now a reduced platinum metal oxide, either alone or as a mixed alloy oxide, is dried thoroughly, such as by the use of a heat lamp, ground, and then sieved through a 400 mesh nylon screen to produce a fine powder of the reduced platinum metal oxide.

The resulting reduced platinum metal oxides are then stabilized thermally by the heating in the presence of oxygen for a sufficient time to ensure a catalytic material which is stable in an scidic hydrogen halide environment and in the presence of halogens. In a manner to be described subsequently, thermal stabilization of the catelyst results in a catalyst which has much better corrosion characteristics in halogens, such as chlorine, etc., and in the presence of halides solutions such a hydrochloric, etc., acids. It is believed that thermal stabilization results in the formation of a catalytic particle having a large mean pore diameter and stable thin oxide film on the outside of the reduced oxide particle. This stabilizes the reduced oxide particles so that they have better mechanical properties for bonding to the solid polymer electrolyte membrane, and in their resistance characteristics to dissolution in hydrochloric acid or other halide acid solutions or

to the evolved halogens. Thus, preferably, the reduced oxides are heated at 350 to 750°C from

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It has also been found that the electrocatalytic activity of the catalyst and of the electrode including the catalyst is optimized by providing the catalytic particles in as fine a powder form as possible. Thus, it has been found that the surface area of the particles, as observed both by the BET nitrogen absorption method, should be at least 25 meters square per gram of catalyst 5 (M<sup>2</sup>/g). The preferred range is 50 to 150M<sup>2</sup>/g.

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The gas permeable electrode structure of catalytic particles and fluorocarbon polymer particles is produced by blending the catalytic particles with a Teflon dispersion to produce a bonded electrode structure in the manner described in U.S. Patent No. 3,297,484 assigned to the assignee of the present invention. In the process of bonding the electrode, it is desirable to 10 blend the catalyst with Teflon dispersions in such a manner that the dispersion contains little or no hydrocarbons. If the fluorocarbon Teflon composition contains hydrocarbon organic surface active agents, it results in loss of surface area of the reduced oxide catalyst. Any reduction on the surface area of the catalyst is obviously undesirable, since it has potentially deleterious effect on the efficiency and effectiveness of the catalyst. Hence, fabrication of the electrode should be 15 by the use of a Teflon polytetrafluoroethylene particle composition which contains few, if any, hydrocarbons. One suitable form of these particles which may be utilized in fabricating the electrode is sold by Dupont under its designation Teflon T-30.

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The mixture of noble metal particles and Teflon particles or of graphite and the reduced oxide particles are placed in a mold and heated until the composition is formed into a decal which is 20 then bonded and embedded in the surface by the application of pressure and heat. As described, for example, in U.S. Patent No. 3,297,484, above, the electrode structure is bonded to the surface of the ion-exchange membrane thus integrally bonding the gas absorbing particle mixture and, in some instances, preferably embedding it into the surface of the membrane.

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The novel membrane/electrode structure thus fabricated comprises a solid polymer electrolyte 25 membrane capable of selective ion transport having a thin, porous, gas permeable electrode of the above-described electrocatalytic reduced platinum group metal oxides bonded to at least one side of the membrane. A second electrode may be bonded to the other side of the membrane and may include the same electrocatalytic material, or any other suitable cathodic material. The selective ion transporting membrane is preferably a stable, hydrated, cationic membrane which 30 is characterized by ion transport selectivity. The cation exchange membrane allows passage of positively charged cations such as hydrogen ions in the case of the electrolysis of a halide such as hydrogen chloride or sodium cations in the case of the electrolysis of aqueous alkali metal halides, and thus minimizes passage of negatively charged anions.

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There are various types of ion exchange resins which may be fabricated into membranes to 35 provide selective transport of the cations. Two classes of such resins are the so-called sulfonic acid cation exchange resins and the carboxylic cation exchange resins. Sulfonic acid exchange resins, which are the preferred type, include ion-exchange groups in the form of hydrated, sulfonic acid radicals (SO<sub>3</sub>H × H<sub>2</sub>O) attached to the polymer backbone by sulfonation. The ion exchanging acid radicals in the membrane are fixedly attached to the backbone of the polymer 40 ensuring that the electrolyte concentration does not vary. As pointed out previously, perfluorocarbon sulfonic acid cation membranes are preferred. One specific class of cation polymer membranes in this category is sold by the Dupont Company under its trade designation "Nafion". The "Nafion" membranes are hydrated, copolymers of polytetrafluoroethylene (PTFE) and polysulfonyl fluoride vinyl ether containing pendant sulfonic acid groups.

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The ion-exchange capacity (IEC) of a given sulfonic cation exchange membrane is dependent upon the milliequivalent weight (MEW) of the SO<sub>3</sub> radical per gram of dry polymer. The greater the concentration of the sulfonic acid radicals, the greater the ion-exchange capacity and hence the capability of the membrane to transport cations. However, as the ion-exchange capacity of the membrane increases, so does the water content and the ability of the membrane to reject 50 salt decreases. Thus in electrolysis of alkali metal halide solutions, caustic is generated at the cathode side and the rate at which the sodium hydroxide migrates from the cathode to the anode side thus increases with IEC. Such back migration reduces the cathodic current efficiency

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(CE) and also results in oxygen generation at the anode which have undesirable consequences in its effect on the catalytic anode electrode. Consequently, the preferred ion-exchange membrane 55 for use in brine electrolysis is a laminate consisting of a thin (2 mil or so) film of fifteen hundred (1500) MEW, low water content (5-15%) cation exchange membrane which has high salt rejection, bonded to a 4 mil or so film of high ion-exchange capacity, 1100 MEW, bonded together with a Teffon cloth. One form of such a laminated construction sold by the Dupont

Company is Nafion 315. Other forms of laminates or constructions in which the cathode side 60 layer consists of a thin layer of film of low water content resin (5-15%) to optimize salt rejection 60 are also available. Typical of such other laminates are Nafion 355, 376, 390, 227, 214. In the case of a laminated membrane bonded together by a Teflon cloth, it may be desirable to clean " !- ----- 1701 agreent HNO for three to four (3

back migration of caustic or other salts, so that simpler forms of membranes such as Nafion 120 may be utilized as the ion transporting medium.

In the case of brine electrolysis, the cathode side barrier layer which must be characterized by low water content may include laminates in which the cathode side layer is a thin, (2-4 mil) 5 chemically modified film of sulfonamide groups or carboxylic acid groups.

Referring now to Fig. 1, the halogen electrolysis cell is shown generally at 10 and consists of a cathode compartment 11 and an anode compartment 12 separated by a solid polymer electrolyte membrane 13 which is preferably a hydrated, permselective, cationic membrane. Bonded to opposite surfaces of membrane 13 are electrodes comprising particles of a 10 fluorocarbon such as Teflon bonded to thermally stabilized, reduced oxides of ruthenium (RuO,) or iridium (IrO,), or stabilized, reduced oxides of ruthenium-iridium (Rulr), ruthenium-titanium (RuTi), ruthenium-tantalum (RuTa), ruthenium-tantalum-iridium (RuTalr), or ruthenium-graphite or combinations of the above with graphite and other valve and transition metal oxides. The cathode, shown at 14, is bonded to and preferably embedded in one side of the membrane and 15 a catalytic anode, not shown, is bonded to and preferably embedded in the opposite side of the membrane. Current collectors in the form of metallic screens 15 and 16 are pressed against the electrodes. The whole membrane/electrode assembly is firmly supported between the housing elements 11 and 12 by means of gaskets 17 and 18 which are made of any material resistant or inert to the cell environment, namely caustic chlorine, oxygen, aqueous sodium chloride in 20 the case of brine electrolysis and HCI, HBr, in the case of other hydrogen halides. One form of such a gasket is a filled rubber gasket sold by Irving Moore Company of Cambridge, Mass. under its trade designation EPDM.

An aqueous alkali metal halide such as brine or hydrogen halides such as HCI is introduced through an electrolyte inlet 19 which communicates with chamber 20. Spent electrolyte and 25 halogens such as chlorine are removed through an outlet conduit 21. A cathode inlet conduit 22 25 is provided in the case of brine electrolysis and communicates with cathode chamber 11 to permit the introduction of the catholyte, water, or aqueous NaOH (more dilute than that formed electrochemically at the electrode/electrolyte interface). In the case of electrolysis of hydrogen halides such as hydrogen chloride, no catholyte need be provided and the cathode inlet conduit 30 22 may be dispensed with.

In a brine electrolysis cell, the water serves two separate functions. A portion of the water is electrolyzed to produce hydroxyl (OH-) anions which combine with the sodium cations transported across the membrane to form caustics (NaOH). The water also sweeps across the porous, bonded cathode electrode to dilute the highly concentrated caustic formed at the 35 membrane/electrode interface to minimize diffusion of the caustic back across the membrane into the anolyte chamber. Cathode outlet conduit 21 communicates with the cathode chamber 11 to remove excess catholyte and the electrolysis products such as caustic in the case of brine electrolysis, plus any hydrogen discharge at the cathode both in brine electrolysis and in hydrogen chloride electrolysis. A power cable 24 is brought into the cathode chamber and a 40 comparable cable not shown is brought into the anode chamber. The cables connect the current 40 conducting screens 15 and 16 or any other suitable kind of collector as source of electrical

Fig. 2 illustrates diagrammatically the reactions taking place in the cell during the electrolysis of an aqueous alkali metal halide such as brine and is useful in understanding the electrolysis 45 process in the manner in which the cell functions. Thus, an aqueous solution of sodium chloride 45 is brought into the anode compartment which is separated from the cathode compartment by the cationic membrane 13. Membrane 13 is a composite membrane comprising a high water content (20-35% based on dry weight of membrane) layer 26, on the anode side and a low water content high MEW cathode side layer, (5-15% based on dry weight of membrane) 50 separated by a Teflon cloth 28. The cathode side barrier layer may also be chemically modified on the cathode side to form a thin layer of a low water content polymer. In one form this is achieved by modifying the polymer to form a substituted sulfonamide membrane layer. By converting the cathode side layer to a weak acid form (sulfonamide), the water content of this portion of the membrane is reduced and the salt rejecting capability of the film is increased. As 55 a result, diffusion of sodium hydroxide back across the membrane to the anode is minimized. While laminated membrane constructions are preferred in brine electrolysis to block migration of sodium hydroxide, other homogeneous films of low water content may be utilized, (viz., Nafion 150, perfluorocarboxylates, etc.). Obviously, in the case of the electrolysis of hydrogen halides such as HCl, HBr, etc., the ion transporting membrane may be a simple, homogeneous film 60 such as the Nafion 120 referred to previously.

The Teflon-bonded, reduced noble metal oxide catalysts contains at least one thermally stabilized, reduced platinum metal oxide, such as ruthenium, iridium, or ruthenium-iridium with or without reduces oxides of titanium. niobium. or tantalum and narticles of aroubite are

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<b>5</b>	provide the electrolyzing pote such as an aqueous sodium of at anode 29 to produce chlorine actually is principally passes through the porous electronsported across membrane 31 is brought into the cathological such as the	to the positive and negative terminals of child across the cell electrodes. The aquichloride solution, is brought into the entrine as shown diagrammatically by the ly evolved at the interface of the electrode actrode to the electrode surface. The solution 13 to cathode 14. A stream of water the chamber and acts as a catholyte. The	peous halide ion solution, ode chamber, is electrolyzed bubble formation 30. The le and the membrane, but dium ions (Na+) are or aqueous NaOH shown at aqueous stream is swept	5
10	armes the surface of the Tell	on-bonded catalytic cathode 14 to dilute and thereby reduce diffusion of the cat	e the caustic formed at the	10
15	A portion of the water cath gaseous hydrogen. The hydromembrane to produce sodium hydroxide readily wets the To surface where it is diluted by Even with a cathode water st	colyte is electrolyzed at the cathode to foxyl ions combine with the sodium ions in hydroxide at the membrane/electrode of the forming part of the bonded electrothe aqueous stream sweeping across the concentrated sodium hydroxide in	transported across the interface. The sodium de and migrates to the ne surface of the electrode. In the range of 4.5–6.5M is	15
20	produced at the cathode. So back through membrane 13 concentration gradient and e bydroxide transported to the	me sodium hydroxide, as shown by the to the enode. NaOH migration is a diffu lectrochemical negative ion transport to anode is oxidized to produce water and	arrow 33, does migrate ision process caused by the the anode. Sodium oxygen as shown by bubble	20
25	efficiency and should be min rejection characteristics on the production of oxygen at the electrode and membrane, pa dilutes the chloring produces	se, is a parasitic reaction which reduces imized by the utilization of membranes he cathode side. Aside from its effect on anode is undesirable since it can have the tricularly if the electrode includes graph at the anode so that processing is required.	which have high saft current efficiency, roublesome effects on the ite. In addition, the oxygen sired to remove the oxygen.	25
30	migrating hydroxide is conve	inimized further by acidifying the aqueo rted to water rather than generating ox or electrolysis of NaCl is as follows:	gen. The reactions in	30
	Anode Reaction: (Principal)	2 Cl→Cl <sub>2</sub> ↑ + 2e <sup>-</sup>	(1)	
35	Membrane Transport: Cathode Reaction:	2Na+ + H₂O 2H₂O→2OH- + H₂↑ - 2e- 2Na+ + 2OH-→2NaOH	(2) (3) (4)	35
	Anode Reaction: Overall	$40H^{-}\rightarrow O_{2} + 2H_{2}O + 4e^{-}$ $2NaCl + 2H_{2}O\rightarrow 2NaOH + Cl_{2}\uparrow + H_{2}\uparrow$	(5) (6)	-
40	(Principal)  The reactions for electrolysis	of a hydrogen halide, such as HCI, are	very similar:	40
			(1)	
Ĭ	Anode Reaction: Membrane Transport:	2HCI→2H+ + Cl₂† + 2e <sup>-</sup> 2H+(H₂O, HCl)	(2)	
45	Cathode Reaction:	2H + + 2e → H₂↑	(3)	45
+0	Overall Reaction:	2HCl→H <sub>2</sub> + Cl <sub>2</sub>	(4)	
50	described herein is character contact with the cation mem backbone (whether these rad carbovylic acid radicals). Con	electrolyzing aqueous solutions of bring ized by the fact that the catalytic sites in brane and the ion exchanging acid radicals are the SO <sub>3</sub> H × H <sub>2</sub> O sulfonic radic sequently, there is no IR drop to speak	n the electrodes are in direct icals attached to the polymer als or the COOH × H₂O of in the anolyte or the	50
	catholyte fluid chambers (this	IR drop is usually referred to as "Elect	trolyte IH drop").	

"Electrolyte IR drop" is characteristic of existing systems and processes in which the electrode 55 and the membrane are separated and can be in the order of 0.2 to 0.5 volts. The elimination or substantial reduction of this voltage drop is, of course, one of the principal advantages of this invention since it has an obvious and very significant effect on the overall cell voltage and the economics of the process. Furthermore, because chlorine is generated directly at the anode and membrane interface, there is no IR drop due to the so-called "bubble effect" which is a gas 60 blending and mass transport loss due to the interruption or blockage of the electrolyte path between the electrode and the membrane. As pointed out previously, in prior art systems, the chlorine discharging catalytic electrode is separated from the membrane. The gas is formed

directly at the electrode and results in a gas layer in the space between the membrane and the

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In a preferred embodiment, the Teflon-bonded noble metal electrode contains reduced oxides of ruthenium, iridium or ruthenium-iridium in order to minimize chloride overvoltage at the anode. The reduced ruthenium oxides are stabilized against chlorine and oxygen evolution to produce an anode which is stable. Stabilization is effected initially by temperature stabilization; i.e., by heating the reduced oxides of ruthenium for one hour at temperatures in the range of 5 550 to 600°C. The Teflon-bonded reduced oxides of ruthenium anode is further stabilized by mixing it with graphite and/or alloying or mixing with reduced oxides of iridium (Ir)O, in the range of 5 to 25% of iridium, with 25% being preferred, or with reduced oxides of titanium (Ti)O<sub>2</sub>, with 25-50% of TiO<sub>2</sub> preferred. It has also been found that a ternary alloy of reduced 10 oxides of titanium, ruthenium and iridium (Ru, Ir, Ti)O, or tantalum, ruthenium and iridium (Ru, Ir, Ta)O, bonded with Teflon is very effective in producing a stable, long-lived anode. In case of the ternary alloy, the composition is preferably 5% to 25% by weight of reduced oxides of iridium, approximately 50% by weight reduced oxides of ruthenium, and the remainder a transition metal such as titanium. For a binary alloy of reduced oxides of ruthenium and 15 titanium, the preferred amount is 50% by weight of titanium with the remainder ruthenium. 15 Titanium, of course, has the additional advantage of being much less expensive than either ruthenium or iridium, and thus is an effective extender which reduces cost while at the same time stabilizing the electrode in an acid environment and against HCI, chlorine and oxygen evolution. Other transition metals, such as niobium (Nb), tantalum (Ta), zirconium (Zr) or 20 hafnium (Hf) can readily be substituted for Ti in the electrode structures. In addition to transition metals, transition metal carbides, nitrides and sulfides may also be utilized as catalyst extenders. The alloys of the reduced noble metal oxides along with the reduced oxides of titaniùm or other transition metals are blended with Teflon to form a homogeneous mix. The anode Teflon content may be 15 to 50% by weight of the Teflon, although 20 to 30% by weight is 25 preferred. The Teflon is of the type as sold by the DuPont Corporation under its designation T-25 30, although other fluorocarbons may be used with equal facility. Typical noble metal, etc., loadings for the anode are 0.6 mg/cm<sup>2</sup> of the electrode surface with the preferred range being 1-2 mg/cm<sup>2</sup>. The current collector for the anode electrode may be a platinized niobium screen of fine mesh which makes good contact with the electrode surface. Alternatively, an expanded 30 titanium screen coated with ruthenium oxide, iridium oxide, transition metal oxide and mixtures 30 thereof may also be used as an anode collector structure. Yet another anode collector structure may be in the form of a titanium-palladium plate with a platinum clad screen attached to the plate by welding or bonding. The cathode is preferably a bonded mixture of Teflon particles and platinum black with 35 platinum black loading of 0.4 to 4 mg/cm<sup>2</sup>. The cathode electrode, like the anode, is bonded to 35 and embedded in the surface of the cation membrane. The cathode is made quite thin, 2-3 mils or less, and preferably approximately 0.5 mils, is porous and has a low Teflon content. The thickness of the cathode can be quite significant in that it can be reflected in reduced water or equeous NaOH sweeping and penetration of the cathode and thus reduces cathodic 40 current efficiency. Cells were constructed with thin (approximately .5 to 2.0 mil) Pt black-15% 40 Teflon bonded cathodes. The current efficiencies of thin cathode cells were approximately 80% at 5M NaOH when operated at 88-91°C with a 290g/L NaCl anode feed. With a 3.0 mil Ru-graphite cathode, the current efficiency was reduced to 54% at 5M NaOH. Table A shows

TABLE A

45 the best performance.

50	Cell	Cathode	Cathode Thickness (mil)	Current Efficiency % (M NaOH)	50
	1	Pt Black	. 2-3	64 (4.0 M)	
	2	Pt Black	2-3	73 (4.5 M)	
	3	Pt Black	1-2	75 (3.1 M)	
55	4	Pt Black	1-2	82 (5 M)	55
	5	Pt Black	0.5	78 (5.5 M)	
	6	5% Pt Black on Graphite	3	78 (3.0 M)	•
60	7	15% Ru O, on Graphite	3	54 (5.0 M)	60
	8	Platinized Graphite Cloth	10–15	57 (5 M)	00

the relationship of CE to thickness, and indicates that thicknesses not exceeding 2-3 mils give

interface to escape readily. It is made porous to allow penetration of the sweep water to the cathode electrode/membrane interface where the NaOH is formed and to allow brine feedstock ready access to the membrane and the electrode catalytic sites. The former aids in diluting the highly concentrated NaOH when initially formed before the NaOH wets the Teflon and rises to 5 the electrode surface to be further diluted by water sweeping across the electrode surface. It is 5 important to dilute at the membrane interface where the NaOH concentration is the greatest. In order to maximize water penetration at the cathode, the Teflon content should not exceed 15% to 30% weight, as Teflon is hydrophobic. With good porosity, a limited Teflon content, a thin cross-section, and a water or diluted caustic sweep, the NaOH concentration is controlled to 10 10 reduce migration of NaOH across the membrane. The current collector for the cathode must be carefully selected since the highly corrosive caustic present at the cathode attacks many materials, especially during shutdown. The current collector may take the form of a nickel screen since nickel is resistant to caustic. Alternatively, the current collector may be constructed of a stainless steel plate with a stainless steel screen 15 welded to the plate. Another cathode current structure which is resistant to or inert in the 15 caustic solution is graphite or graphite in combination with a nickel screen pressed to the plate and against the surface of the electrode. **EXAMPLES** Cells incorporating ion exchange membranes having Teflon-bonded reduced noble metal oxide 20 electrodes embedded in the membrane were built and tested to illustrate the effect of various parameters on the effectiveness of the cell in brine electrolysis and to illustrate particularly the operating voltage characteristics of the cell. Table I illustrates the effect on cell voltage of the various combinations of the reduced noble 25 metal oxides. Cells were constructed with electrodes containing various specific combinations of reduced noble metal oxides bonded to Teflon particles and embedded into a cationic membrane 6 mils thick. The cell was operated with a current density of 300 amperes per square foot at 90°C, at feed rates of 200 to 2000 cc per minutes, with feed concentration of 5M. One cell was constructed in accordance with the teachings of the prior art and contained a 30 30 dimensionally stabilized anode spaced from the membrane and a stainless steel cathode screen similarly spaced. This control cell was operated under the same conditions. It can readily be observed from this data that in the process of the instant invention, the cell

It can readily be observed from this data that in the process of the instant invention, the cell operating potentials are in the range of 2.9–3.6 volts. When compared to a typical prior art arrangement (Control Cell No. 4), under the same operating conditions, a voltage improvement of 0.6V-1.5V is realized. The operating efficiencies and economic benefits which result are

clearly apparent.

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Cell No.	Anode	Cathode	Brine Feed	Current Density (ASF)	Cell Voltage (V)	1c	ы ы	Membrane (5M NaOH)
-	6 Mg/Cm <sup>2</sup> ·	4 Mg/Cm²	~5M	300	3.2-3.3	06	85%	DuPont Nafion 315
7	6 Mg/Cm²	rt black 4 Mg/Cm² 8: 6101	(7/8067) "	300	3.3-3.6	.06	78%	DuPont 1500 EW
က	6 Mg/Cm²	, ,	ŧ	300	2.9	.06	%99	DuPont 1500 Ew
4	Dimensionally Stable Screen Anode—Spaced	Stainless Steel Screen Spaced		300	4.2-4.4	.06	81%	DuPont 1500 EW Nation
ß	4 Mg/Cm²	from Memorane 4 Mg/Cm² 6- biock	:	300	3.6-3.7	.06	85%	DuPont Nation 315
9		4 Mg/Cm <sup>2</sup>	:	300	3.5-3.6	.06	%98	DuPont Nation 315
7	6 Mg/Cm²	Pt Black 2 Mg/Cm²	:	300	3.0	.06	88%	DuPont Nafion 315
80	6 Mg/Cm²	4 Mg/Cm²	:	300	3.4	80.	83%	DuPont 1500 EW
6	6 Mg/Cm²	4 Mg/Cm²	:	300	3.4-3.7	.06	73%	DuPont 1500 EW
10	2 Mg/cm²	4 Mg/Cm <sup>2</sup>		300	3.1-3.5	.06	80%	DuPont Nation 315
-	2 Mg/Cm² (Ir Ox)	4 Mg/Cm² Pt Black	:	300	3.2-3.6	.06	65%	DuPont Nafion 315 Laminate

A cell similar to Cell No. 7 of Table I was constructed and operated at 90°C in a saturated brine feed. The cell potential (V) as a function of current density (ASF) was observed and is shown in Table II.

## 5 TABLE II

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	Cell Voltage (V)	Current Density (ASF)
	3.2	400
10	2.9	300
••	2.7	200
	2.4	100

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This data shows that call operating potential is reduced as current density is reduced. Current density vs. cell voltage is, however, a trade-off between operating and capital costs of a chlorine electrolysis. It is significant, however, that even at very high current densities (300 and 400 ASF), significant improvements (in the order of a volt or more) in cell voltages are realized in the chlorine generating process of the instant invention.

Table III illustrates, the effect of cathodic current efficiency on oxygen evolution. A cell having Tellon-bonded reduced noble metal oxides catalytic anodes and cathodes embedded in a cationic membrane were operated at 90°C with a saturated brine concentration, with a current density of 300 ASF and a feed rate of 2–5 cc/Min/in² of electrode area. The volume percent of oxygen in the chlorine was determined as a function of cathodic current efficiency.

25 TABLE III 25

Cathodic Current Efficiency (%)	Oxygen Evolution (Volume %)
89	2.2
86	4.0
84	5.8
80	<b>8.9</b> .

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Table IV illustrates the controlling effect that acidifying the brine has on oxygen evolution. The volume percent of oxygen in the chlorine was measured for various concentration of HCI in the brine.

TABLE IV

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45	Acid (HCI) Concentration (M)	Oxygen Volume %
45	0.05	2.5
	0.75	1.5
	0.10	0.9
	0.15	0.5
50	0.25	0.4

It is clear from this data that oxygen evolution due to electrochemical oxidation of the back migrating OH<sup>-</sup> is reduced by preferentially reacting the OH<sup>-</sup> chemically with H<sup>+</sup> to form H<sub>2</sub>O.

A cell similar to Cell No. 1 of Table I was constructed and operated with a saturated NaCl feedstock acidified with 0.2M HCl and at 300 ASF. The cell voltage was measured at various operating temperatures from 35–90°C.

A cell similar to Cell No. 7 of Table I was constructed and operated with 290g/L (~5M)/L NaCl stock (not acidified) at 200 ASF. The cell voltage was measured at various operating 60 temperatures from 35-90°C. The data was normalized for 300 ASF.

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	TABLE V			-
5	Cell No. 1 Voltage	Cell No. 7 Voltage Normalized to 300 ASF (200 ASF Data)	Temperature *C	
	3.65 3.38	3.50 (3.15) 3.30 (2.98)	35° 45°	-
	3.2	3.20 (2.9)	55*	
10	3.15	3.12 (2.78)	65°	
	3.10	3.05 (2.72)	75°	
	3.05 3.02	2.97 (2.65) 2.95 (2.63)	85°	
	3.02	2.55 (2.05)		_
15				•
20	surface of an ion trai improved process pe electrodes is bonded improvement in such	nsporting membrane, the may informance is achieved for all to the surface of the ion tren	kimum improven structures in whi sporting membe hat less than is t	ich at least one of the or (hybrid cell). The the case with both electrodes
25	voltage requirements A number of cells a fully bonded cell (bonded and cathode	s for known processes). were constructed and brine e both electrodes) with the resu only bonded) and with the re	lectrolysis carried ts in hybrid cell esults a prior art	d out to compare the results in constructions (anode only
30	cell was operated at electrode catalyst los RuO <sub>s</sub> -graphite and R	90°C with a brine feedstock adings were 2g/ft <sup>2</sup> at the cath uO <sub>x</sub> . The current efficiency at	of approximately node for Pt Black : 300 ASF was e	
35	TABLE VI			
				Cell Voltage (V)

(	Cell	Anode	Cathode	Cell Voltage (V) at 300 ASF	. •
40	1	Ru-Graphite (Bonded)	Pt Black (Bonded)	2.9	40
	2	Platinized Niobium Screen (Not Bonded)	Pt Black (Bonded)	3.5	•
45	3	Platinized Niobium Screen (Not Bonded)	Pt Black (Bonded)	3.4	45
•	4	Ru-Graphite (Bonded)	Ni Screen (Not Bonded)	3.5	_
	5	RuO, (Bonded)	Ni Screen (Not Bonded)	3.3	_
50 (	6	Platinized Niobium Screen (Not Bonded)	Ni Screen (Not Bonded)	3.8	50_

It can be seen that the cell voltage of the fully Teflon-bonded cell No. 1 is almost a volt better than the voltage for the prior art, completely non-Teflon bonded, control cell No. 6. Hybrid cathode bonded cells 2 and 3 and hybrid anode bonded cells 4 and 5 are approximately 0.4–0.6 volts worse than the fully Teflon-bonded cell but still 0.3–0.5 volts better than the prior art processes which are carried out in a cell without any Teflon bonded electrodes.

It will be appreciated that a vastly superior process for generating chlorine and other halides
from brine and, as will be shown hereafter, from HCI and other halides, has been made possible
by reacting the analyte and the catholyte at catalytic electrodes bonded directly to and
embedded in the cationic membrane. By virtue of this arrangement, the catalytic sites in the
electrodes are in direct contact with the membrane and the acid exchanging radicals in the

fluorocarbon bonded thermally stabilized, reduced noble metal oxide catalysts, as well as fluorocarbon graphite-reduced noble metal oxide catalysts with low overvoltages, further enhance the efficiency of the process.

_	EYA		21	
	EYA	ΛÆ	M	2.5

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Electrodes containing thermally stabilized, reduced noble metal oxides, etc., embedded in ion-exchange membranes were built and tested to illustrate the effect of various parameters on the effectiveness of the cell and catalyst in the electrolysis of hydrochloric acid.

Table VII illustrates the Effect on Cell Voltage of various combinations of reduced noble metal oxides. Cells were constructed with Teflon-bonded, graphite electrodes containing various specific combinations of thermally stabilized, reduced platinum metal oxides and reduced oxides of titanium embedded into a hydrated cationic membrane, 12 mils thick. The cell was operated with a current density of 400 amps per square, at 30°C, at a feed rate of 70 cc per minutes, (0.05 ft² active cell area) with feed normalities of 9–11N.

Table VIII and IX illustrate the effect of time for the same cells and under the same conditions, on cell operating voltages.

Table X shows the effect of acid feed concentration ranging from 7.5–10.5N. A cell, like cell No. 5 in Table II, was constructed with reduced (Ru, 25% Ir)O, noble metals added to the Teflon-bonded graphite electrode. The cell was operated at fixed feed rate of 150 cc/min, (0.05 to 20 ft² active cell area) at 30°C and 400 ASF.

TABLE VII

No.	Operational Time (Hrs.)	Anode- Graphite/ Fluorocarbon Plus	Loading Mg/Cm²	Cathode Graphite/ Fluorocarbon Plus	Loading (Mg/Cm²)	Feed Normality (Eq/L)	Current Density— Amperes/Sq. Ft. (ASF)	Cell Voltages (V)
-	6300	(Ru)Ox Heat Stabilized	9.0	(Ru)Ox	9.0	9-11	400	2.10
N	6300	(Ru Ti)Ox Heat Stabilized	9.0	(Ru Ti)Ox	9.0	:	400	2.01
က	4900	(Ru Ti)Ox Heat Stabilized	1.0	(Ru Ti)Ox	1.0	:	400	1.97
4	1800		1.0	(Ru)Ox	1.0	:	400	1.91
LΩ	4000	(Ru 25% Ir)Ox	1.0	(Ru 26% Ir)Ox	1.0	:	400	2.07
9	200	(Ru, Ti, 5% Ir)Ox Heat Stabilized	2.0	(Ru, Ti, 5% Ir)Ox	2.0	:	400	1.80
7	100	(Ru-25% Ta)Ox	2.2	(Ru, 25% Ta)Ox	2.0	:	400	1.64
		. 0000						

\*Performance of this cell at 3800 hours was approximately 1.9V. Taken off test due to cell leakage

13					GB 2 009 788A	13
	TABL	E VIII				•
5	Cell No.	Cell Voltage (V) at 100 Hrs. Operating Time	Cell Voltage (V) At Operating Time From Table I	Current Density Amperes Per Square Foot (ASF)		5
10	3 4	1.85 1.84 1.78 1.80	2.10 2.01 1.97 1.91	400 400 400 400 400		10
, 15	<ul><li>5</li><li>6</li></ul>	1.75 1.70	2.07* (1.9) 1.80	400		15
		note for Table VII	•		•	
20	TABL	E <i>IX</i> .			·	20
25	Cell No.	Intermediate Operating Time—(Hrs.)	Current Density Amperes/Sq. Ft.(ASF)	Cell Voltages (V)		<sub>.</sub> 25
	1	3900	100 200 300	1.70 1.93 2.00		20
30	2	3400	100 200 300	1.57 1.70 1.83		30
35	3	1900	100 200 300	1.58 1.70 1.81		35
40	4	1000	100 200 300	1.47 1.60 1.72		40
45	5	1200	100 200 300	1.32 1.45 1.55		45
•	TABL	E X				50
. 50	Feed (Eq/L	Normality }	Volume %			
55	7 7.5 8 8.5		0.4 0.15 0.04 0.015	·		55
60	10 10.5 11.5		0.007 0.004 0.003			60

voltage and low temperature (~30°C) operation resulting in economical operation of such electrolyzer cells. Furthermore, this data shows excellent performance at various current densities, particularly at 300–400 ASF. This has a positive and beneficial effect on capital costs for chlorine electrolyzers embodying the instant invention.

To show the effect of thermal stabilization on reduced noble metal and transition metal oxides, certain tests were carried out. These tests show the impact on the resistance of the catalyst to harsh electrolysis environments. Thermally stabilized as well as non-stabilized, reduced oxide catalysts were exposed to highly concentrated HCl solutions which represent extremely harsh environmental conditions. The color of the solution was observed since darkening of the solution indicated loss of catalyst. Increasing loss of catalyst was accompanied by more pronounced color changes.

Table XI shows the results of these corrosion resistance and stability tests for catalyst batches

ranging from .5 to 20 gms.

Catalyst	Treatment	Temp. *C	Corrosion Medium	Time (Hours)	Observation (Color)	Stability Evaluation
Ru O,	None	24.C	12N HCI	24	Light Brown Color	Modest Corrosion
	Thermally Stabilized	24.C	12N HCI	744	Very Pale Yellow	Very Little Corrosion Good Stability
(Ru 25Nb)O,	None	24°C	12N HCI	24 912	Light Brown . Amber	Modest Corrosion
(Ru 50Ta)O.	None	24°C	12N HCI	168	Pale Amber	Modest Corrosion
	550°C for one (1) Hour 550°C for one (1) More	24.C	12N HCI	96 72	Very Pale Yellow No Change in	Fully Stable
(Ru 5ir)O.	Hour	24.C	12N HCI	168	Color J Amber	Substantial Corrosion
	550°C for one (1) Hour	. 24°C	12N HCI	96	No Change in Color	Custable Fully Stable
(Ru 251r)O,	None	24°C	12N HCI	168	Amber	Substantial corrosion Unstable
	650°C for one (1) Hour 650°C for one (1) More Hour	24.C	12N HCI 12N HCI	96	Very Pale Yellow No Color Change	Fully Stable

TABLE XI

It will be obvious from this data that thermal stabilization of the reduced oxides enhances the corrosion resistance of the catalyst in very concentrated HCl and, in fact, provides very good stability. It is obvious that resistance of the catalysts in the much less corrosive chlorine or brine environments is excellent and attributable to thermal stabilization of the reduced oxide catalyst. Having observed the improved corrosion characteristics of the thermally stabilized, reduced, 5 platinum group metal oxides, physical and chemical tests were conducted to determine the effect of thermal stabilization on various characteristics of the catalysts which might account for the improved corrosion characteristics. The oxide content, surface area in M2/g of catalyst, the pore volume and pore size distribution of the catalyst were measured after fabrication of the 10 catalyst by the modified Adams method; after reduction of the catalyst; and after thermal 101 stabilization of the reduced catalyst. The result of these tests, which will be set forth in detail below, show that the surface area of the catalyst is reduced somewhat after the catalyst is reduced, and quite substantially after thermal stabilization. A drop in oxide content after the reduction step is believed to account for part of the decrease of the surface area. A substantial 15 change in the internal pore size distribution of the catalyst after thermal stabilization without a 15 corresponding change in the poor volume is believed responsible for the very substantial decrease (ratio of 2 to 1) in surface area accompanying thermal stabilization and would account for the improved corrosion characteristics as corrosion is directly related to the area exposed to attack by any corrosive agents. Initially, Sample #1, a ruthenium-25% by weight iridium catalyst was prepared by the 20 . modified Adams method. A portion of this catalyst was reduced electrochemically to form Sample #2. A reduced (Ru 25 Ir)O, sample was thermally stabilized for one (1) hour at 550-600°C. The surface area of the unreduced (Sample #1) catalyst, the reduced catalyst (Sample #2) and the thermally stabilized, reduced (Ru 25 Ir)O, catalyst (Sample #3), as 25 measured by the three point BET (BRUNAUER-EMMET-TELLER) nitrogen adsorption method, is 25 shown in Table XII. TABLE XII 30 Catalyst 30 (Ru 25 lr) **Treatment** Surface Area Sample #1 None 127.6 M<sup>2</sup>/g Sample #2 Reduced 123.5 M<sup>2</sup>/g 35 Sample #3 Reduced and 62.3 M<sup>2</sup>/g 35 Thermal Stabilization-550-600°C; One (1) Hour 40 40

The oxide content of Samples #1, #2, and #3 was then measured as well as that of a Sample (#4) thermally stabilized at 700-750°C for one (1) hour. In addition the oxide content of Pt Ir catalysts containing respectively 5 and 50% by weight of Iridium was measured. The results are shown in Table XIII.

Decrease of oxide content (i.e., unreduced catalyst) will have a corresponding effect of surface area since the surface area of oxides is normally greater than that of the non-oxide for This reduction in oxide content in part explains surface area reduction but does not wholly explain the dramatic reduction in surface area after thermal stabilization.  The prescipt of the patches was therefore measured to determine whether thermal stabilization.	nole #2 Reduction Thermal Stabilization; 550–600°C—One (1) Hour Die #4 Reduction and Thermal Stabilization; 700–750°C— One (1) Hour	24.3 22.6
Sample #3  Thermal Stabilization; 550–600°C—One (1) Hour  Sample #4  Reduction and Thermal Stabilization; 700–750°C— One (1) Hour  (Pt-50Ir)  Sample #5  None 16.5  Sample #6  Reduction Stabilization; 550–600°C— One (1) Hour  The data from Table XIII shows a decrease in oxide content with reduction and thermal stabilization just as surface area decreases after reduction and thermal stabilization.  Decrease of oxide content (i.e., unreduced catalyst) will have a corresponding effect on surface area since the surface area of oxides is normally greater than that of the non-oxide for this reduction in oxide content in part explains surface area reduction but does not wholly explain the dramatic reduction in surface area after thermal stabilization.	ole #3  Thermal Stabilization; 550–600°C—One (1) Hour Neduction and Thermal Stabilization; 700–750°C— One (1) Hour	22.6
550–600°C—One (1) Hour  Sample #4 Reduction and Thermal 21.5 Stabilization; 700–750°C— One (1) Hour  (Pt-50Ir) Sample #5 None 16.5 Sample #6 Reduction 15.2 Sample #7 Reduction and Thermal 13.0 Stabilization; 550–600°C— One (1) Hour  The data from Table XIII shows a decrease in oxide content with reduction and thermal stabilization just as surface area decreases after reduction and thermal stabilization.  Decrease of oxide content (i.e., unreduced catalyst) will have a corresponding effect on surface area since the surface area of oxides is normally greater than that of the non-oxide for this reduction in oxide content in part explains surface area reduction but does not wholly explain the dramatic reduction in surface area after thermal stabilization.	550-600°COne (1) Hour Reduction and Thermal Stabilization; 700-750°C One (1) Hour	•
Sample #4 Reduction and Thermal 21.5  Stabilization; 700-750°C— One (1) Hour  (Pt-50Ir)  Sample #5 None 16.5  Sample #6 Reduction 15.2  Sample #7 Reduction and Thermal 13.0  Stabilization; 550-600°C— One (1) Hour  The data from Table XIII shows a decrease in oxide content with reduction and thermal stabilization just as surface area decreases after reduction and thermal stabilization.  Decrease of oxide content (i.e., unreduced catalyst) will have a corresponding effect on surface area since the surface area of oxides is normally greater than that of the non-oxide for This reduction in oxide content in part explains surface area reduction but does not wholly explain the dramatic reduction in surface area after thermal stabilization.  The nearest the of the certalyst was therefore measured to determine whether thermal stabilization.	ole #4 Reduction and Thermal Stabilization; 700-750°C One (1) Hour	21.5
Sample #5 None 16.5 Sample #6 Reduction 15.2 Sample #7 Reduction and Thermal 13.0 Stabilization; 550-600°C— One (1) Hour  The data from Table XIII shows a decrease in oxide content with reduction and thermal stabilization just as surface area decreases after reduction and thermal stabilization.  Decrease of oxide content (i.e., unreduced catalyst) will have a corresponding effect on surface area since the surface area of oxides is normally greater than that of the non-oxide for this reduction in oxide content in part explains surface area reduction but does not wholly explain the dramatic reduction in surface area after thermal stabilization.		
Sample #6 Reduction and Thermal 13.0  Stabilization; 550-600°C— One (1) Hour  The data from Table XIII shows a decrease in oxide content with reduction and thermal stabilization just as surface area decreases after reduction and thermal stabilization.  Decrease of oxide content (i.e., unreduced catalyst) will have a corresponding effect on surface area since the surface area of oxides is normally greater than that of the non-oxide for this reduction in oxide content in part explains surface area reduction but does not wholly explain the dramatic reduction in surface area after thermal stabilization.	de #E Noss	
Sample #7  Reduction and Thermal  Stabilization;  550-600°C—  One (1) Hour  The data from Table XIII shows a decrease in oxide content with reduction and thermal stabilization just as surface area decreases after reduction and thermal stabilization.  Decrease of oxide content (i.e., unreduced catalyst) will have a corresponding effect on surface area since the surface area of oxides is normally greater than that of the non-oxide for this reduction in oxide content in part explains surface area reduction but does not wholly explain the dramatic reduction in surface area after thermal stabilization.  The perceive of the establish was therefore measured to determine whether thermal stabilization.	pio ii e	, <del>, , , , , , , , , , , , , , , , , , </del>
Stabilization; 550-600°C— One (1) Hour  The data from Table XIII shows a decrease in oxide content with reduction and thermal stabilization just as surface area decreases after reduction and thermal stabilization.  Decrease of oxide content (i.e., unreduced catalyst) will have a corresponding effect on surface area since the surface area of oxides is normally greater than that of the non-oxide for this reduction in oxide content in part explains surface area reduction but does not wholly explain the dramatic reduction in surface area after thermal stabilization.	, , , , , , , , , , , , , , , , , , , ,	
The data from Table XIII shows a decrease in oxide content with reduction and thermal stabilization just as surface area decreases after reduction and thermal stabilization.  Decrease of oxide content (i.e., unreduced catalyst) will have a corresponding effect on surface area since the surface area of oxides is normally greater than that of the non-oxide for this reduction in oxide content in part explains surface area reduction but does not wholly explain the dramatic reduction in surface area after thermal stabilization.	Stabilization; 550-600°C	13.0
of the catalyst causes a change in porosity thereby decreasing the surface and increasing corrosion resistance. Catalyst samples were taken from the same batches as Samples #1, #2 and the perceity of the samples and particle size distribution measured. Particle size	reduction in oxide content in part explains to the dramatic reduction in surface area at e porosity of the catalyst was therefore mean actalyst acceptant there are a change in porosity there	urface area reduction but does not wholly er thermal stabilization. cured to determine whether thermal stabilization of decreasing the surface area and increasing
distribution was measured by a sedimentation method and showed that the equivalent spher diameter at 50% mass distribution was 3.7 micros (μ) after reduction of the catalyst and 3.1 microns (μ) after thermal stabilization. This indicates that the external surface of the particles reduced but again does not account for all of the surface area reduction after stabilization. Total pore volume data (cc/g) was obtained by capillary condensation and mercury intrusi methods. Data for Samples #1, #2, and #3 is shown in Table XIV.	sion resistance. Catalyst samples were take	cle size distribution measured. Particle size
TABLE XIV	ision resistance. Catalyst samples were take #3 and the porosity of the samples and parabution was measured by a sedimentation mater at 50% mass distribution was 3.7 mic ons (µ) after thermal stabilization. This indicated but again does not account for all of the tell note volume data (cc/o) was obtained be	cle size distribution measured. Particle size of those and showed that the equivalent spherical (a) after reduction of the catalyst and 3.1 tes that the external surface of the particles is surface area reduction after stabilization.  capillary condensation and mercury intrusion
Catalyst Total Pore Volume	ision resistance. Catalyst samples were take #3 and the porosity of the samples and par bution was measured by a sedimentation meter at 50% mass distribution was 3.7 mic ons $(\mu)$ after thermal stabilization. This indicated but again does not account for all of that pore volume data (cc/g) was obtained be tods. Data for Samples #1, #2, and #3 is a sequence.	cle size distribution measured. Particle size of those and showed that the equivalent spherical (a) after reduction of the catalyst and 3.1 tes that the external surface of the particles is surface area reduction after stabilization.  capillary condensation and mercury intrusion
	sion resistance. Catalyst samples were take #3 and the porosity of the samples and par bution was measured by a sedimentation π leter at 50% mass distribution was 3.7 mic ons (μ) after thermal stabilization. This indic ced but again does not account for all of the tall pore volume data (cc/g) was obtained be lods. Data for Samples #1, #2, and #3 is a LE XIV	cle size distribution measured. Particle size othor and showed that the equivalent spherical (µ) after reduction of the catalyst and 3.1 tes that the external surface of the particles is surface area reduction after stabilization. capillary condensation and mercury intrusion nown in Table XIV.
(ma-20 m) Troduction	sion resistance. Catalyst samples were take #3 and the porosity of the samples and par bution was measured by a sedimentation π leter at 50% mass distribution was 3.7 mic ons (μ) after thermal stabilization. This indic ced but again does not account for all of the tall pore volume data (cc/g) was obtained be lods. Data for Samples #1, #2, and #3 is a LE XIV	cle size distribution measured. Particle size othod and showed that the equivalent sphericals (µ) after reduction of the catalyst and 3.1 tes that the external surface of the particles is surface area reduction after stabilization. capillary condensation and mercury intrusion nown in Table XIV.
0.00 /	sion resistance. Catalyst samples were take #3 and the porosity of the samples and par bution was measured by a sedimentation π leter at 50% mass distribution was 3.7 mic ons (μ) after thermal stabilization. This indicced but again does not account for all of the tall pore volume data (cc/g) was obtained be lods. Data for Samples #1, #2, and #3 is set LE XIV	cle size distribution measured. Particle size of size distribution measured. Particle size of the dand showed that the equivalent spherical size (µ) after reduction of the catalyst and 3.1 test that the external surface of the particles is surface area reduction after stabilization. capillary condensation and mercury intrusion nown in Table XIV.  Total Pore Volume cc/g
(110-20 11) (1021110111	sion resistance. Catalyst samples were take #3 and the porosity of the samples and par bution was measured by a sedimentation π leter at 50% mass distribution was 3.7 mic ons (μ) after thermal stabilization. This indicced but again does not account for all of the tall pore volume data (cc/g) was obtained be lods. Data for Samples #1, #2, and #3 is set LE XIV  (lyst	cle size distribution measured. Particle size of size distribution measured. Particle size of the dand showed that the equivalent spherical size (µ) after reduction of the catalyst and 3.1 test that the external surface of the particles is surface area reduction after stabilization. capillary condensation and mercury intrusion nown in Table XIV.  Total Pore Volume cc/g  0.80 cc/gm

The data shows that the total pore volume is relatively unchanged. Thus the porosity, in terms of gms/cc or if converted to void volume (knowing the spherical size and density), is essentially the same and is in the range of 0.7-0.8 gms/cc for Ru-25 Ir.

Thermal Stabilization:

500-600°C-

One (1) Hour

50

Simultaneously, the pore size distribution was measured to obtain the pore diameter distributions in the 40Å-10 micron range. Capillary condensation was used in the 40–500Å range. In this method liquid condensation for a given vapor pressure is measured to obtain pore 60 size distribution. The capillary condensation method has a lower resolution limit of 40Å and an upper limit of 500Å. For pores in excess of 500Å (i.e., 500Å-10μ) a mercury intrusion method is utilized to obtain pore size distribution. Pore diameter distribution measurements for the two ranges is shown in Tables XV and XVI respectively.

TABLE XV

			· · · · · · · · · · · · · · · · · · ·		
5	Catalyst	Treatment	Size Range	Pore Diameter Distribution	5
Ð	Sample #1	None	40-500A*	Pore Distribution below 40A*	. •
10	Sample #2	Reduction	"	Pore diameter distribution below 40A*	10
	Sample #3	Reduction and Thermal Stabilization; 550-600°C; One (1) Hour	<b>,,</b>	Distribution in the range of 100–300A° with maximum at 200A°	15
15					10
	TABLE XVI				
20		~		Pore Diameter Distribution (50% point	20
	Catalyst	Treatment	Size Range	in distribution)	
25	Sample #1 Sample #2 Sample #3	None Reduction Reduction and Thermal Stabilization; 500-600°C;	500A*-10 microns 	0.46μ 0.82μ 1.5μ	25
30		One (1) Hour		•	30
	diameter. This overall surface diameter distri seems quite cl pores. Above surface area the	change seems to be accepted and decreases. With the bution changing so that the dear that many of the pore 500A° the pore diameter has changes with thermal	ompanied by a change a pore volume staying some staying some shows a set below 40A* coalesce increases. The pore size stabilization. In summer	results in a change in pore size in the number of pores so that the substantially constant and the pore a maximum at 200A*, and $1.5\mu$ it is, reducing the overall number of the diameter and internal pore ary, the internal porosity and thus	<b>35</b>
40	result of chang diameter distri distribution to	ging the morphology to p ibution. With the pore vol a larger pore diameter, it	rovide a smaller numbe lume relatively unchang t seems clear that the s	abilized. It is believed that this is a er of pores with a larger pore ged and a change in pore diamete surface area reduction associated change in internal pore surface	
45	area. The porosity stabilized, redupreferred poro	(in terms of pore volume uced platinum metal oxid sity for a thermally stabili	e (cc) per unit weight (g e catalyst lies in the O. ized Ru-25 Ir catalyst l	g) of catalyst) of the thermally 4–1.5 cc/gm range with the	45 ,
50	distribution be the pore diame distribution is distribution).	low 500A° in the 100–3 eter distribution shows th in 0.4–9 $\mu$ range with a n	IOOA* range, with a ma e greatest pore volume naximum at 1.5µ's (rep	aximum at 200A*. Above 500A* and hence the principal presenting the 50% point is	50,
55	should be such without transit corrosion) for t	surface area range for th	ermally stabilized, redu	iced platinum metal oxide catalyst	
	10M²/am ran	ion metals, etc., the surfa the required catalytic acti	ace area should be as lovity. Thus, the surface	tion of platinum metals with or ow as possible (to reduce area should be in excess of being from 24–165M <sup>2</sup> /g and	s 55

	CLAIMS  1. A catalyst system including at least one electroconductive, thermally stabilized, reduced	
	-letinum aroun motel Avide	
	2. The catalyst according to Claim 1 wherein the platinum group metal oxide incided	5
5	therein is ruthenium oxide or iridium oxide.  3. The catalyst according to Claims 1 or 2 which includes electroconductive graphite in	
	t distriction and the abound by etablicant platinum afalla MATRI OXIGO.	
	4. The catalyst according to Claims 1-3 which includes thermally stabilized oxides of a	
10	transition metal.  5. The catalyst according to Claims 2-4 wherein the platinum group metal oxide is	10
U	nuthenium oxide and which includes at least one additional thermally stabilized, reduced metallic	
	auide et icidium tentalum titanium niobium, zirconium or namium.	
	6. The catalyst according to Claim 5 which includes 5% to 25% by weight of reduced oxides of iridium.	
15	7. The catalyst according to Claims 5 or 6 which includes 25% by weight of reduced oxides	15
•	_£ initalia ann	
	8. The catalyst according to Claims 5-7 which includes reduced oxides of iridium and either tantalum or titanium.	
	The catalyst according to Claims 2–4 wherein the platinum group metal oxide is indiam	20
20	and and which includes reduced oxides of tantalum of Manium.	20
	10. The catalyst according to Claims 5–9 which further includes electroconductive graphite.  11. A halogen evolving catalyst including at least one electroconductive, reduced platinum.	
	metal avida sharmally stabilized by heating at an elevated temperature from 500-700 C.	
	12. The catalyst according to Claim 11 with the surface area thereof being between	25
25	24-185M²/gm of catalyst.  13. A porous catalyst including at least one electroconductive, thermally stabilized, reduced	
	missions aroun metal axide having a surface area in excess of Tulvi-/gitt, tile porosity of solo	
	catalyst being from 0.4-1.5cc/gm of catalyst, the pore diameter distribution having been	
30	modified by thermal stabilization to reduce internal pore surface area.  14. The catalyst according to Claim 13 wherein the pore diameter distribution has maxima	30
<b>J</b> U	contared at 2004' and a 50% nore distribution point at 1.0%.	
	15. A catalyst including at least one electroconductive, thermally stabilized, reduced platinum group metal oxide, the surface area of the catalyst being at least 60 meters per gram	
	-AA-hist	٥.
35	16 A combination electrists and electrode structure comprising an ion transporting mem-	35
	brane having at least one gas permeable catalytic electrode bonded to a surface of the membrane, said electrode comprising thermally stabilized, electroconductive, reduced oxides of	
	et least one nistinum group metal.	
	47 The electrolists and electrode structure according to Claim 16 Wherein the 985	40
40	permeable electrody comprises a plurality of thermally stabilized, electroconductive particles of	
	reduced platinum group metal oxides.  18. The electrolyte and electrode structure according to Claim 17 wherein said particles	
	include the evidence of at least two metals of the group consisting of thermally stabilized,	
4-	electroconductive, reduced platinum group metal oxides and reduced transition metal oxides with at least one kind being a reduced platinum group metal oxide.	45
45	10 The electrolyte and electrode structure according to Claim 10 wherein said particles	
	include thermally stabilized electroconductive, reduced 0xIdes of rumenium.	
	20. The electrolyte and electrode structure according to Claims 18 or 19 wherein said particles include thermally stabilized, electroconductive, reduced oxides of ruthenium and at particles include thermally stabilized, electroconductive, reduced oxides of ruthenium and at	
50	least one reduced metallic oxide chosen from the reduced oxides of iridium, tantalum, titanium,	50
	-iahium ziraaaium ar hafaium	
	21. The electrolyte and electrode structure according to Claim 20 wherein the thermally stabilized, electroconductive particles are reduced oxides of ruthenium and reduced oxides of	
	iridium ·	55
55	22. The electrolyte and electrode structure according to Claim 21 wherein the particles	<b>J</b> J
	include 5% to 25% by weight of reduced oxides of iridium.  23. The electrolyte and electrode structure according to Claims 20 or 21 wherein the	
	andicles include 25% by weight of itidilim.	
^-	24 The electrolities and electrode structure according to Claims 18-21 wherein the pivramy	60
ĎÜ	of particles include thermally stabilized, electroconductive particles of reduced noble metal oxides and reduced oxides of titanium or tantalum.	
	25. The electrolyte and electrode according to Claim 24 wherein the reduced hobie metal	
	avida is radiusad iridium avida	

î

	27. A combination electrolyte and electrode structure for halogens comprising an ion transporting membrane having at least one gas permeable electrode bonded to one surface of said membrane, said electrode comprising thermally stabilized, electroconductive, catalytically active reduced oxide particles of at least one platinum group metal, the surface area of the electrode being at least 60 meters per gram of particles.  28. A catalytic electrode structure comprising an agglomerate of electroconductive catalytic particles bonded together with particles of a resinous binder, said catalytic particles including thermally stabilized, reduced oxides of at least one platinum group metal, said electrode structure being gas permeable, electronically conductive and catalytically active for electrolysis	5
10	of halides.  29. The catalytic electrode structure according to Claim 28 wherein said particles include the oxides of at least two metals of the group consisting of electroconductive, thermally stabilized, reduced platinum group metal oxides and thermally stabilized, reduced transition	10
15	metal oxides, with at least one of said oxides being a reduced platinum group metal oxide.  30. The catalytic electrode structure according to Claims 28 or 29 wherein said catalytic particles are thermally stabilized, reduced oxides of ruthenium.	15
20	31. The catalytic electrode structure according to Claims 28–30 which further includes particles of another thermally stabilized, reduced oxides of a platinum group metal.  32. A process for the preparation of the electrocatalytic material comprising forming an oxide of at least one platinum group metal optionally containing a valve metal oxide, reducing the said oxides to a partially oxidized state, heating the reduced oxides in the presence of oxygen at a temperature and for a duration sufficiently to stabilize said reduced oxides, and	20
25	optionally adding up to 50% by weight of graphite.  33. The process according to Claim 32 wherein the reduced oxides are heated at 300 to 750°C.	25
	<ul> <li>34. The process according to Claim 33 wherein the reduced oxides are heated at 550 to 600°C.</li> <li>35. A catalytic material substantially as hereinbefore described.</li> <li>36. An electrode structure substantially as hereinbefore described with reference to the</li> </ul>	
30	accompanying drawings.  37. An electrolytic cell substantially in accordance with any of the Examples of the invention hereinbefore described.	30
	38. A process for preparing an electrocatalytic material, substantially as hereinbefore described.	

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